

*Computer simulation aspects of
nanoparticle and nanodevice design*

Panagiotis Grammatikopoulos

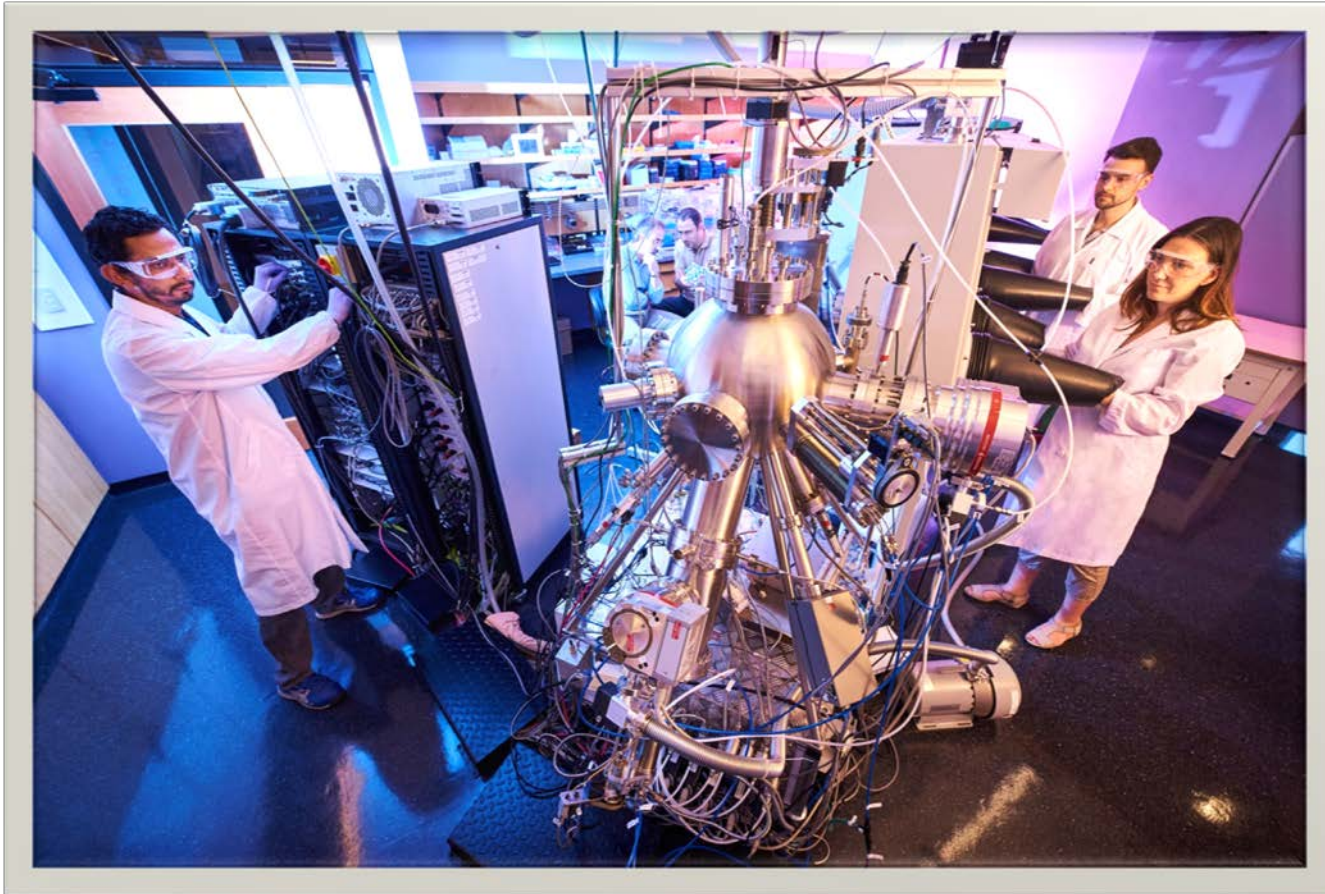


26 January 2021

OIST's *Nanoparticles by Design* Unit



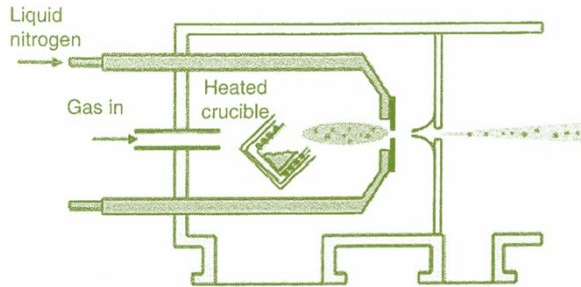
OIST's *Nanoparticles by Design* Unit



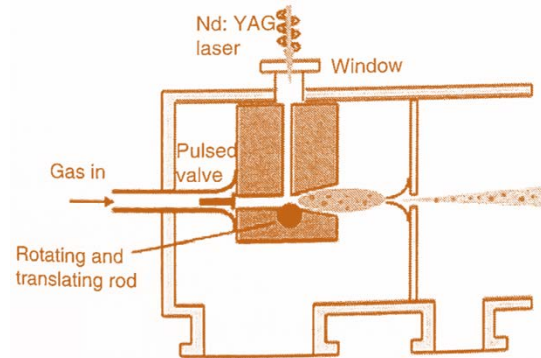
These labs are remarkably diverse – here's why they're winning at science, Nature 2018

Cluster Beam Deposition Sources

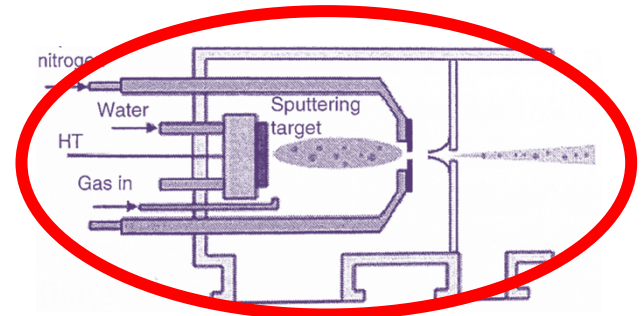
using rare gas to produce supersaturated vapours



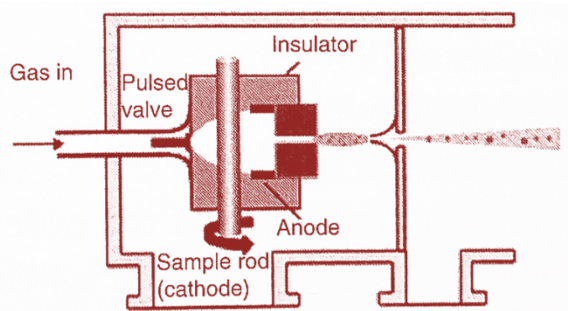
thermal gas aggregation



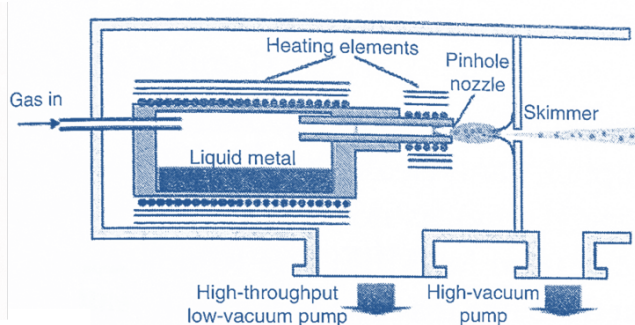
laser ablation



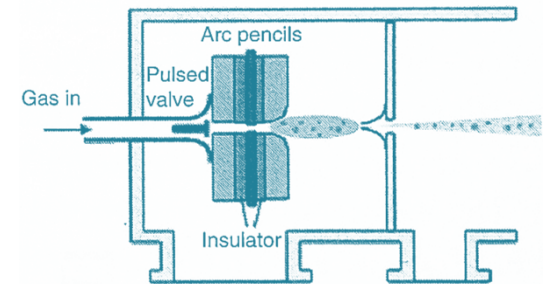
sputter gas aggregation



pulsed microplasma

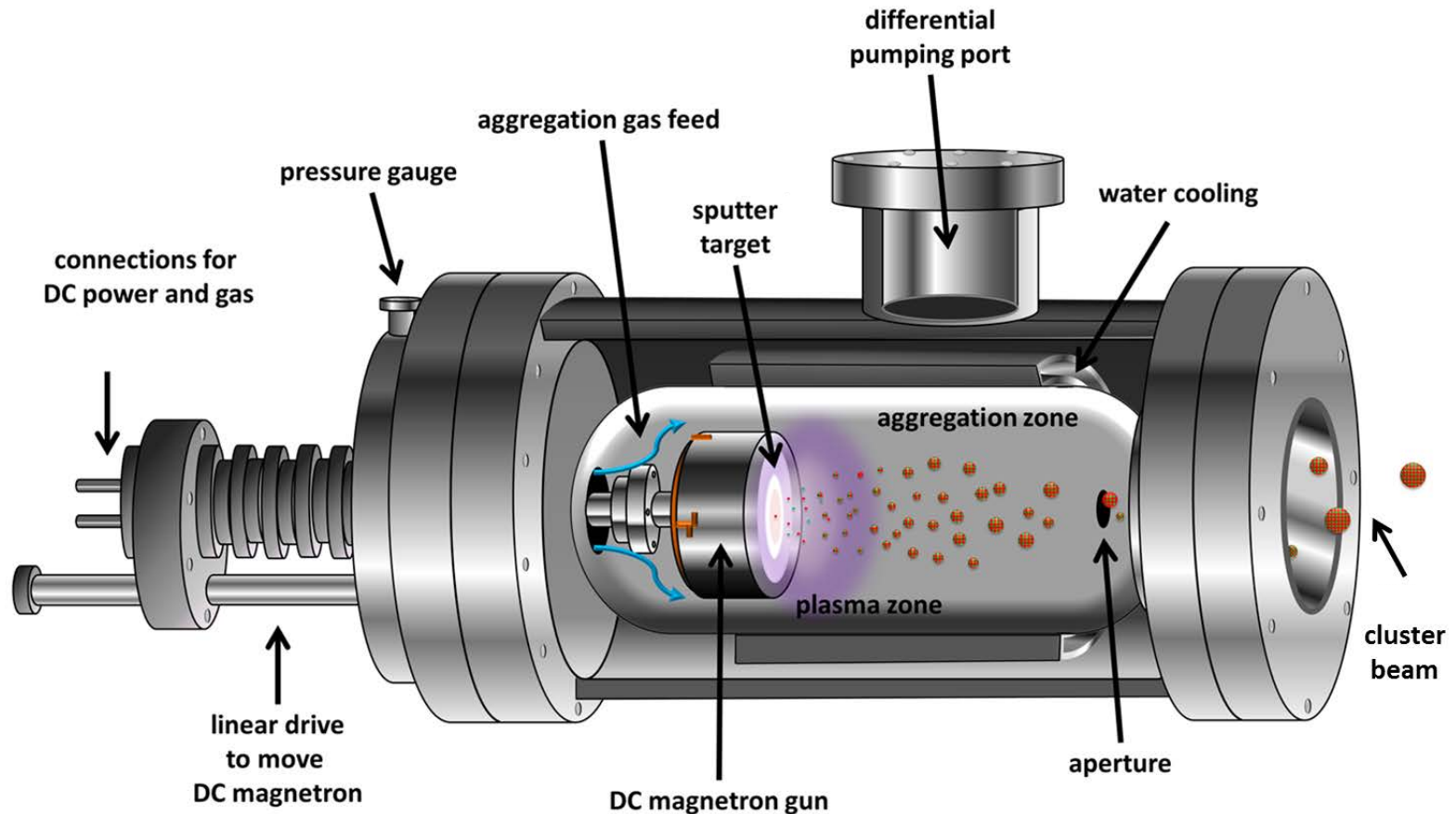


seeded supersonic nozzle



pulsed-arc cluster ion

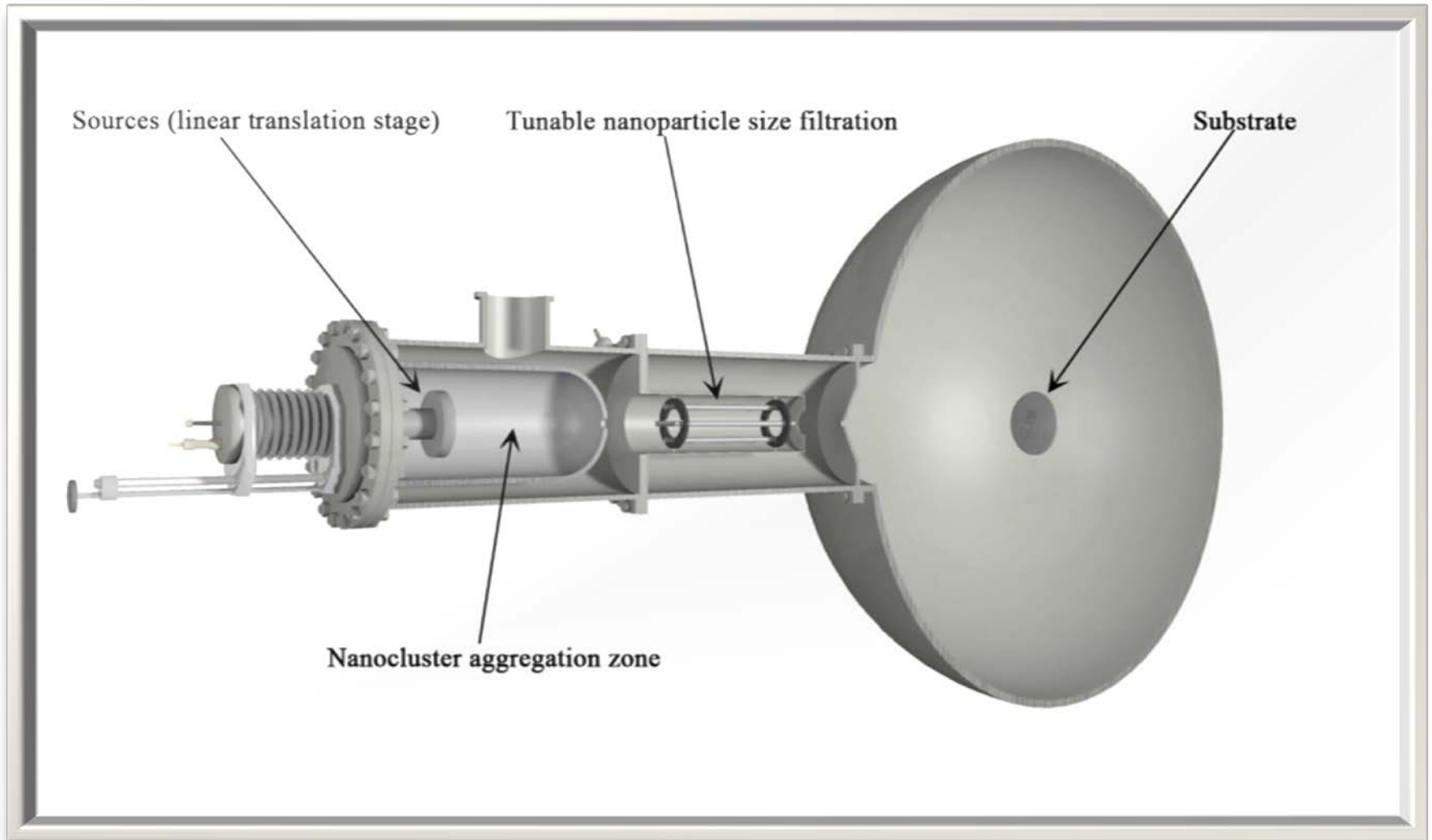
Magnetron-sputtering inert-gas condensation



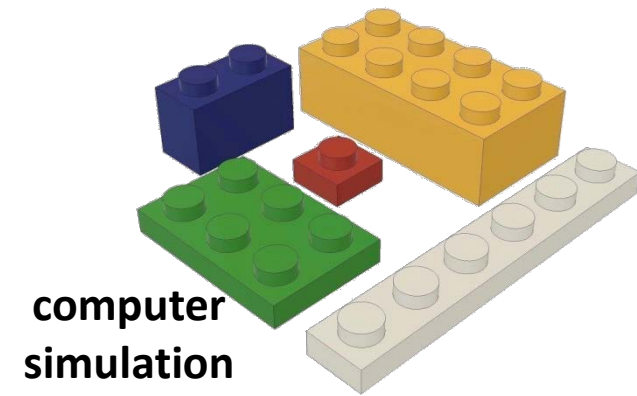
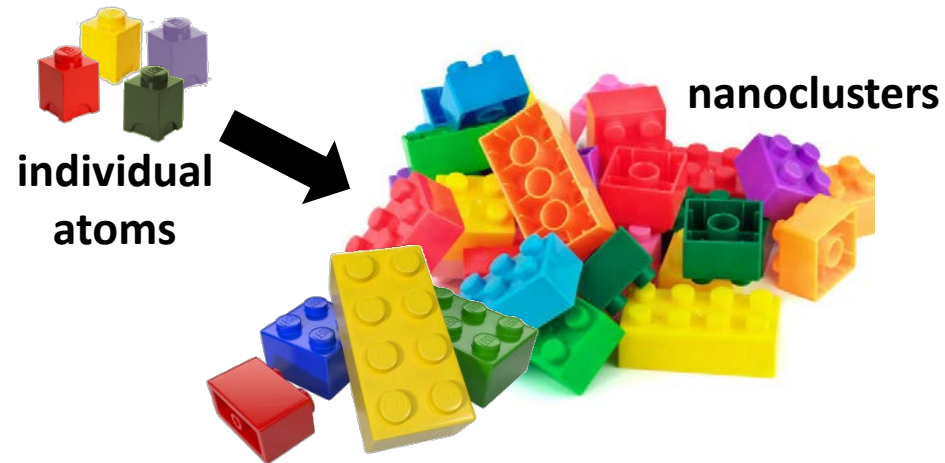
Schematic representation of a DC magnetron-sputtering, inert-gas condensation, cluster beam deposition system, utilizing a single alloy target of the desired composition.

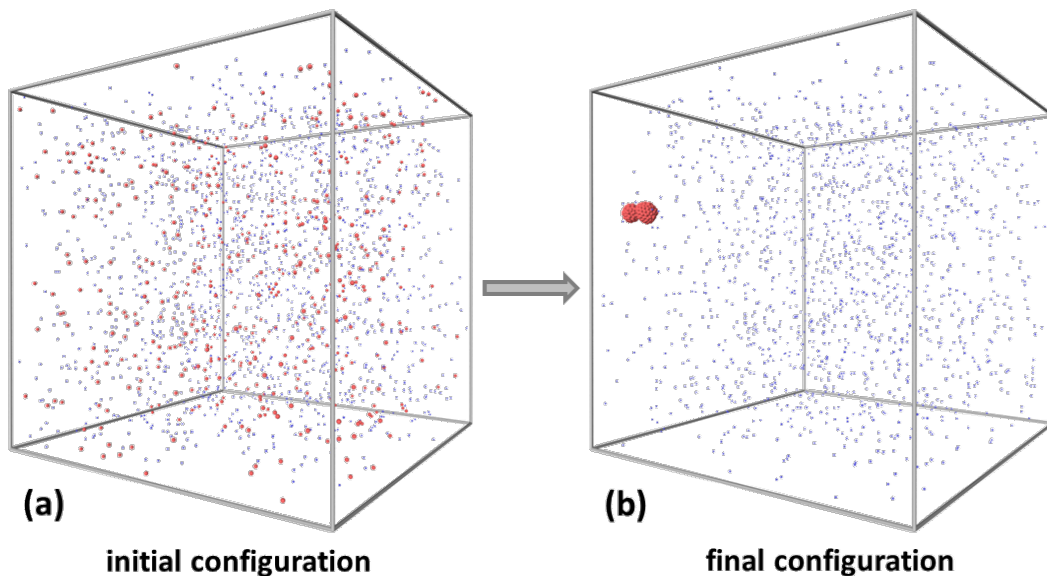
"Tuning the onset of ferromagnetism in heterogeneous bimetallic nanoparticles by gas phase doping"
M. Bohra, P. Grammatikopoulos, V. Singh, J. Zhao, E. Toulkeridou, S. Steinhauer, J. Kioseoglou, J.-F. Bobo, K. Nordlund, F. Djurabekova, M. Sowwan, *Phys Rev Mater* (2017)

Magnetron-sputtering inert-gas condensation



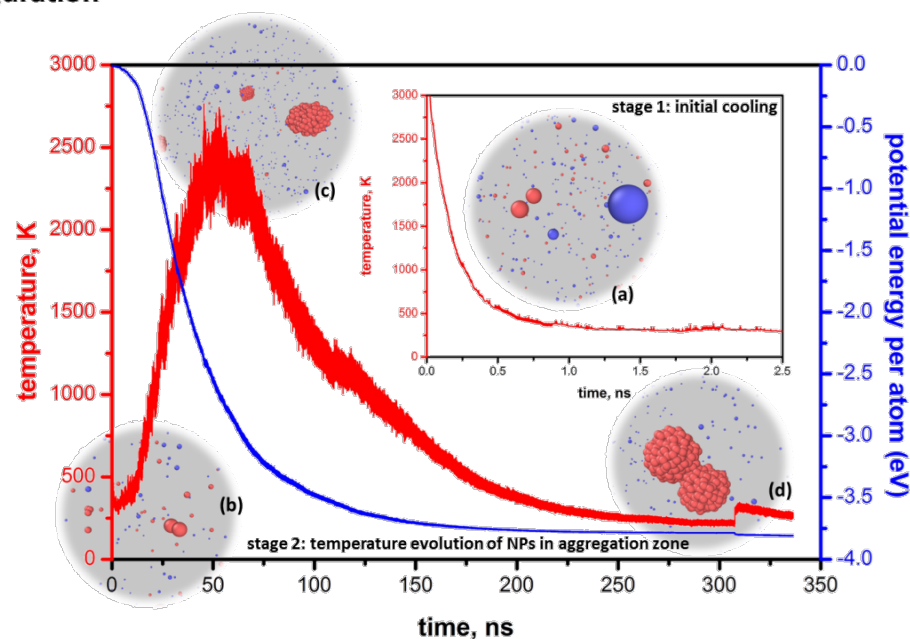
From Nanoparticles by Design...





MD simulation of Si NP growth:
500 Si (red) and 1500 Ar (blue) atoms.

Temperature and potential energy evolution during growth. The inset zooms in at the initial 2.5 ns of the process. MD simulation snapshots designate various stages of the growth mechanism.



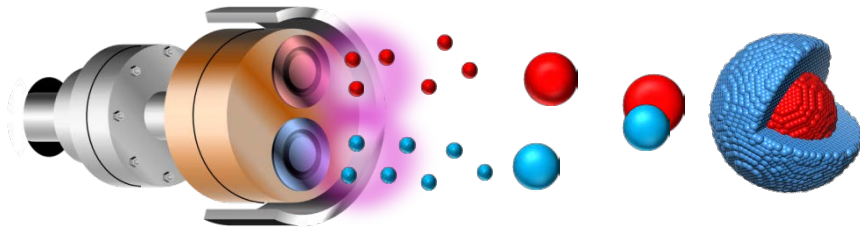
"Nanoparticle formation via magnetron sputtering with inert gas aggregation",
chapter 16 of *"Nanostructured semiconductors: amorphisation and thermal properties"*

P. Grammatikopoulos, M. Sowwan

CRC Press, Boca Raton (2017), ISBN: 978-9-814-74564-2.

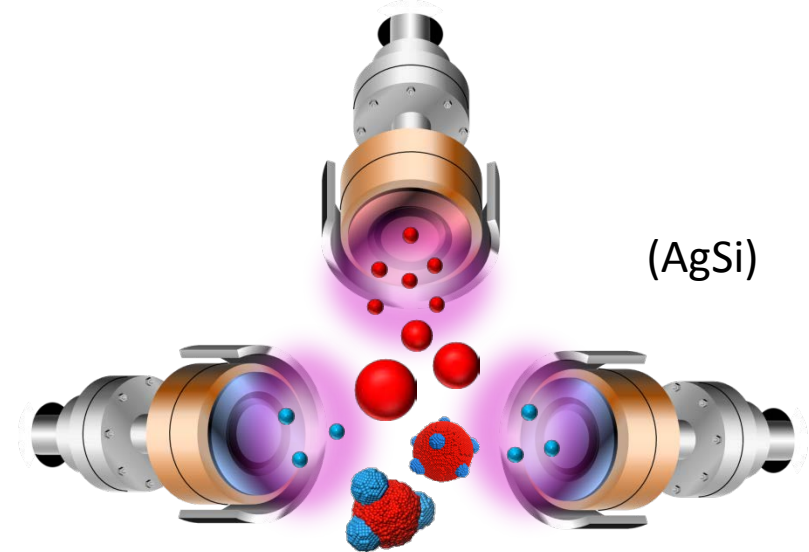
Nanoalloys form via various setups

(i) Two separate targets



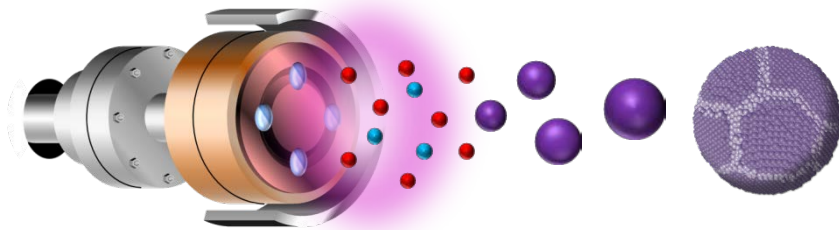
(AgCu, FeAu, FeAl, FePd, AgSi, PdMg, NiPt)

(ii) Post-growth shell coating



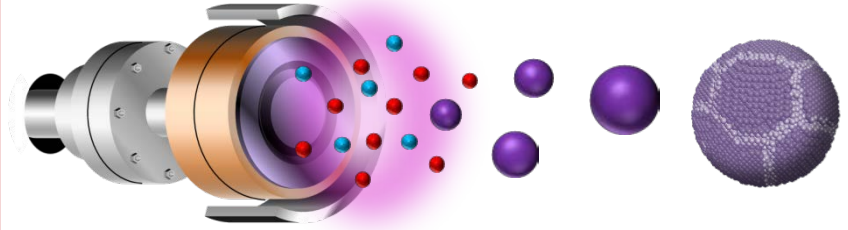
(AgSi)

(iii) One composite target



(FeAu)

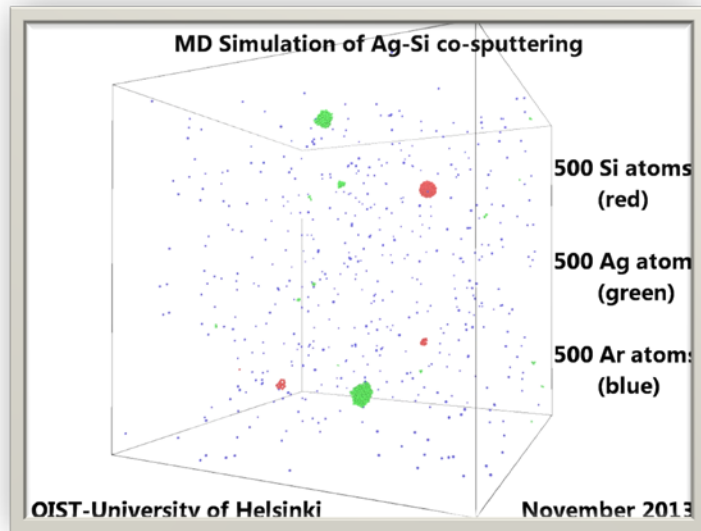
(iv) One alloy target



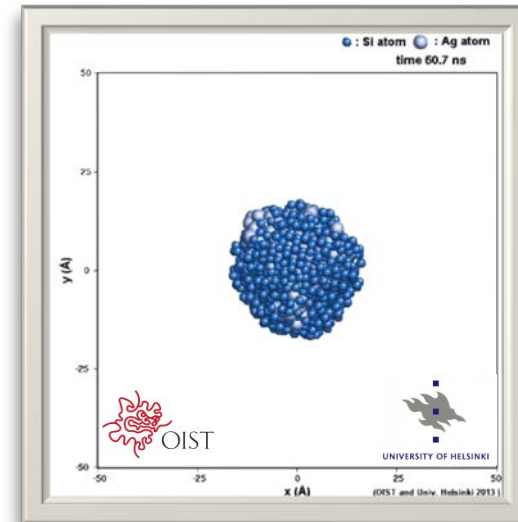
(NiCr)

Nanoalloys form via various setups

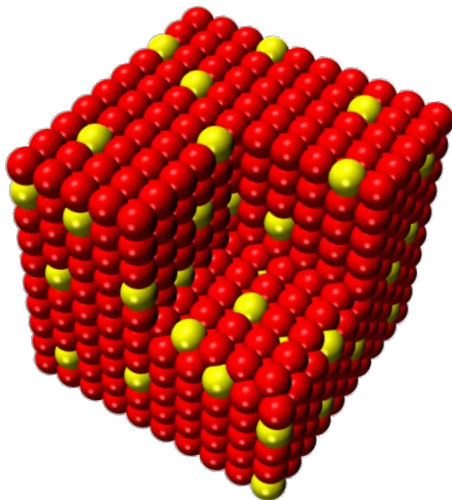
(i) Two separate targets



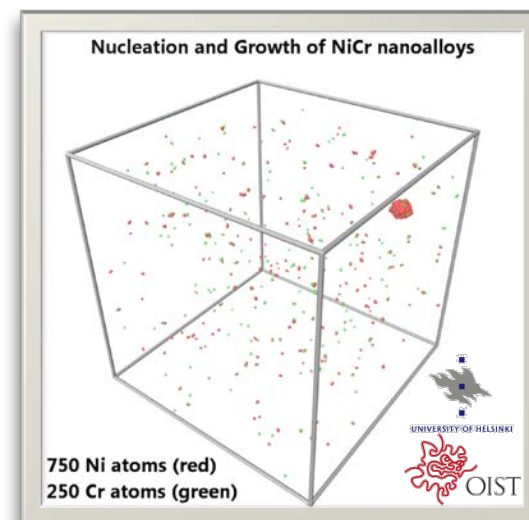
(ii) Post-growth shell coating



(iii) One composite target

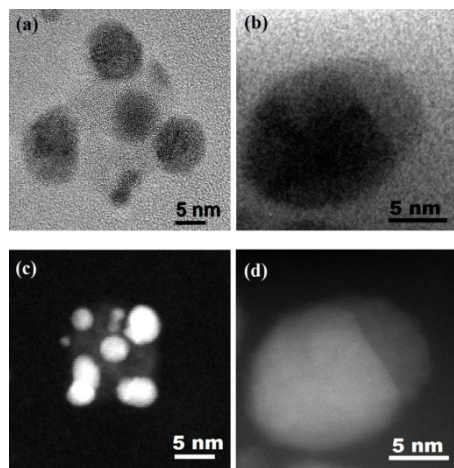


(iv) One alloy target



Nanoalloys form via various setups

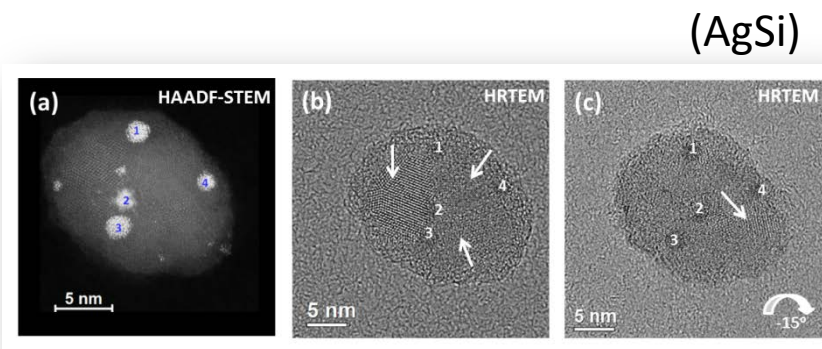
(i) Two separate targets



(AgSi)

Bright-field HRTEM (a, b) and HAADF-STEM (c, d) image of the core-satellite (a, c) and Janus (b, d) Si-Ag nanoparticles, respectively.

(ii) Post-growth shell coating

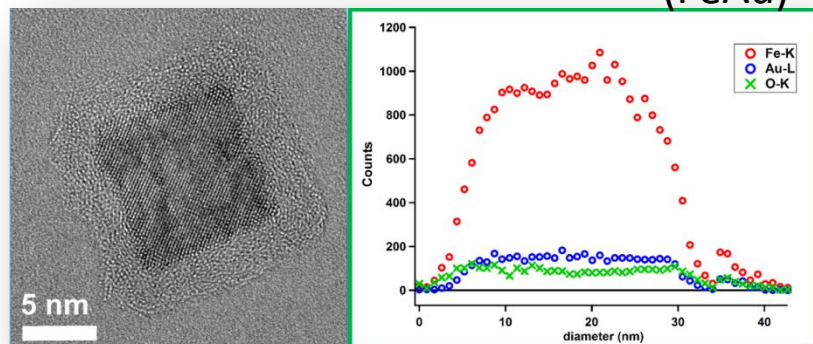


(AgSi)

Ag clusters are numbered 1–4. Three crystal grains are indicated by arrows. (c) A fourth crystal grain is confirmed by tilting the sample by -15° relative to the incident electron beam. Qualitative correlation between the number of Ag clusters and the number of grains in the Si core is evident.

(iii) One composite target

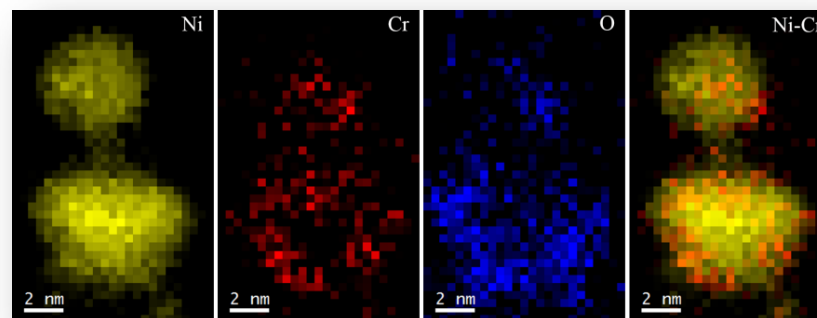
(FeAu)



HR-TEM image and EDX linescan profile of a representative single crystalline FeAu nanocube.

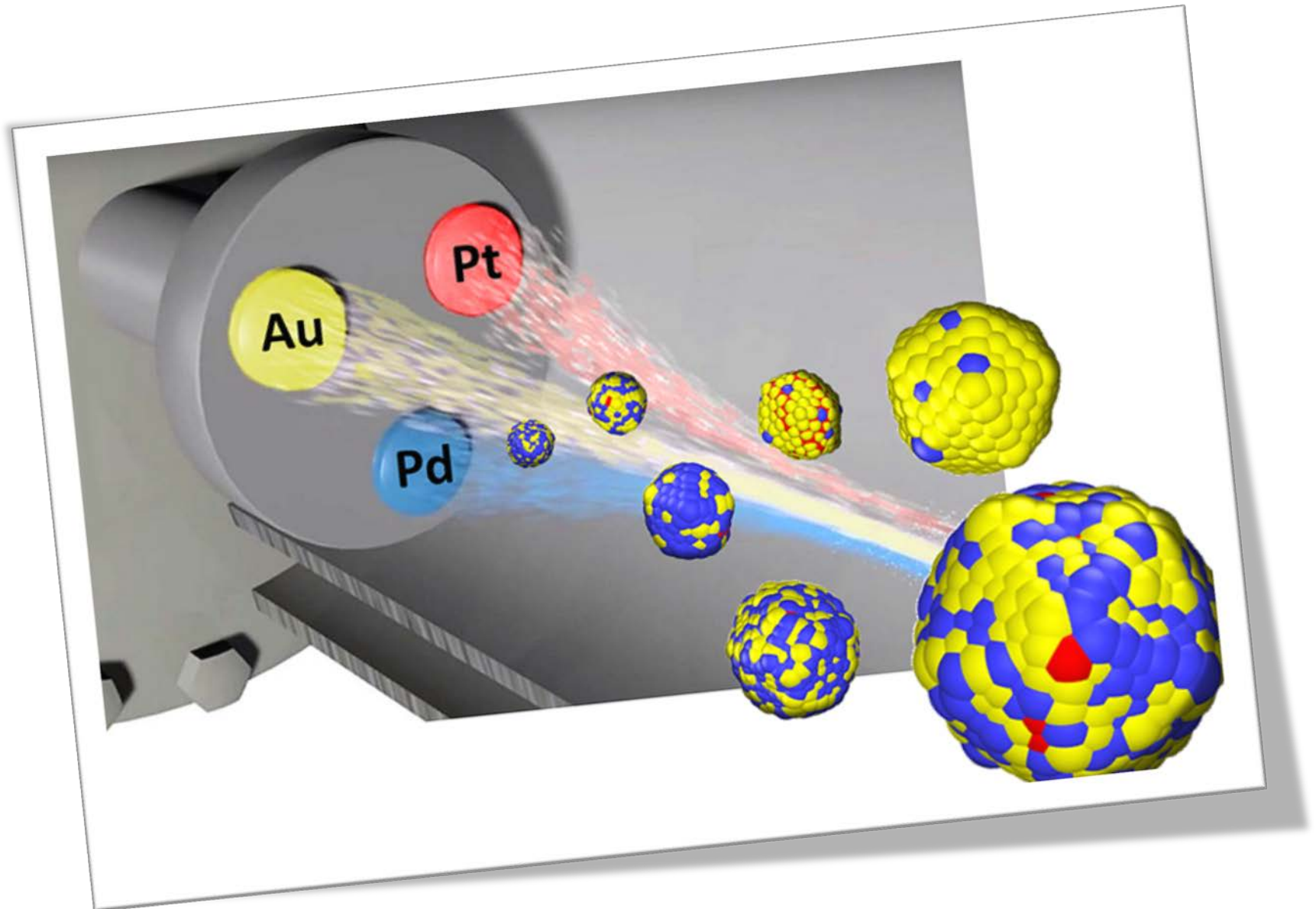
(iv) One alloy target

(NiCr)



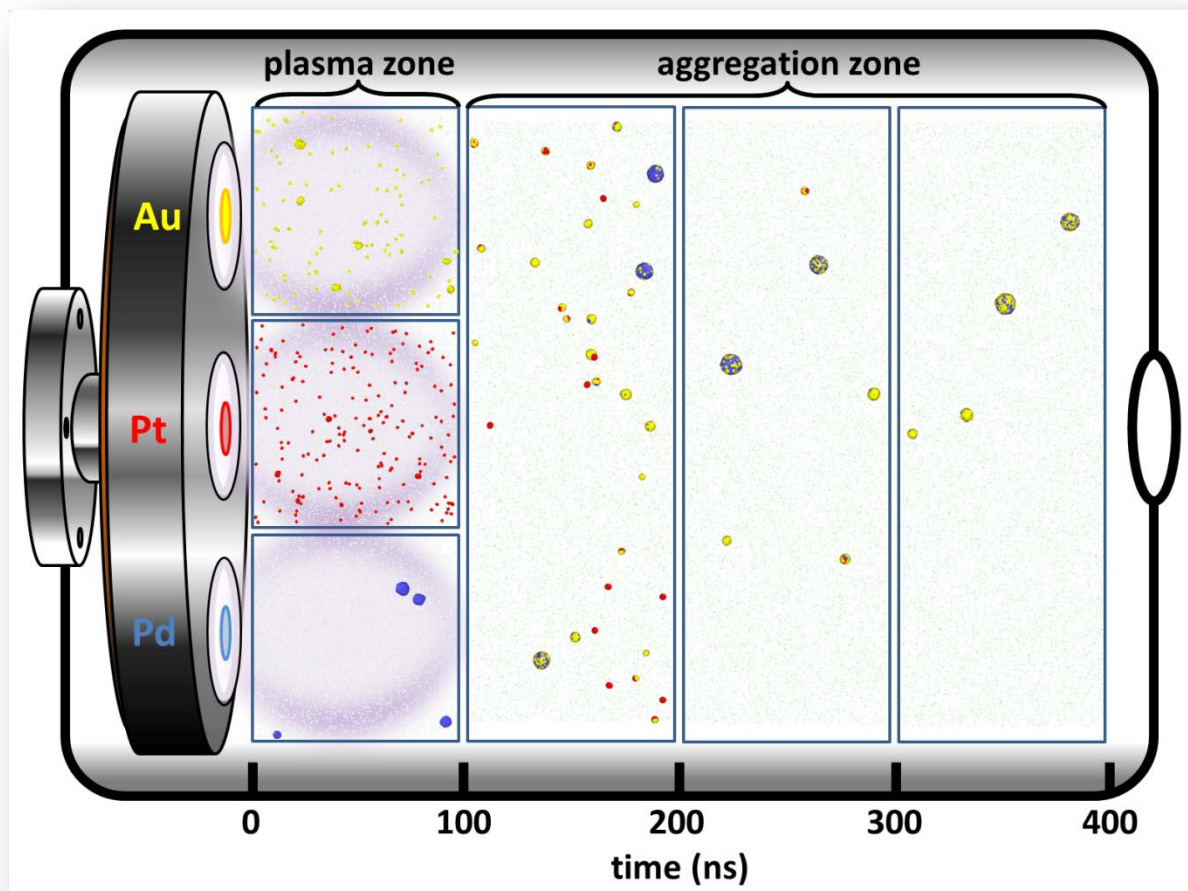
EELS elemental maps of two NCs showing Ni cores and NiCrOx shells. The oxidation is due to exposure to air during transportation of the sample.

Nucleation rate varies with element affecting size of clusters and chemical order of nanoalloys



NP nucleation & growth simulation setup

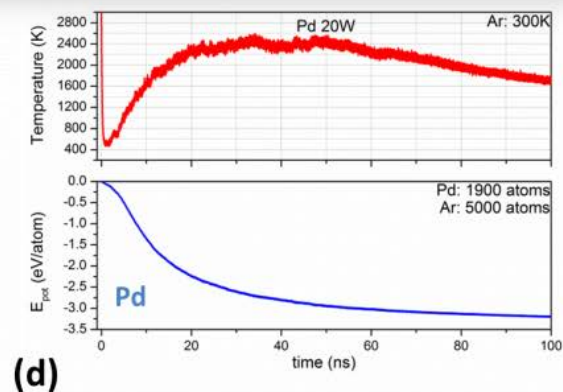
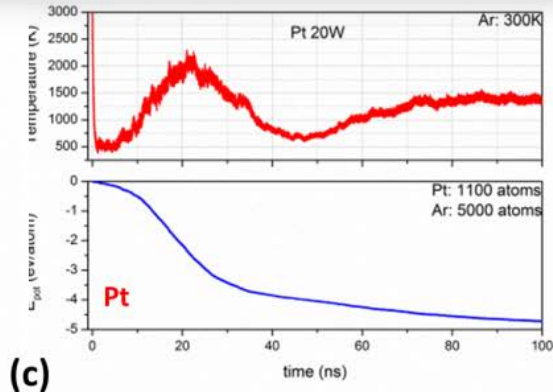
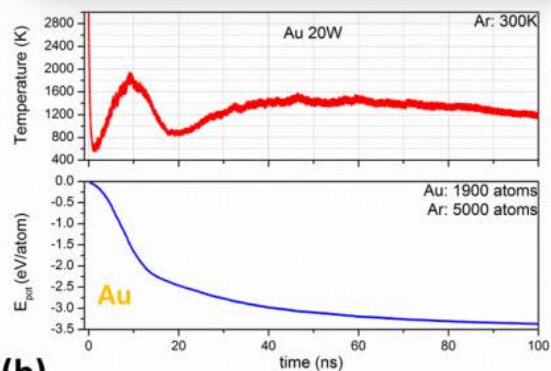
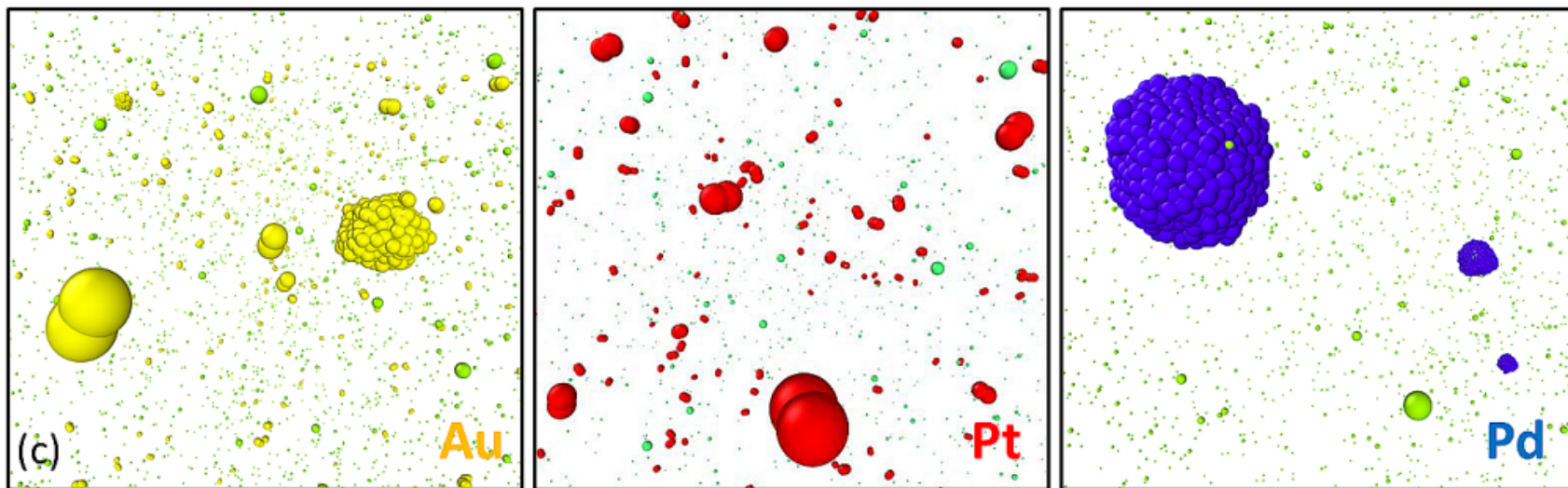
MD growth of trimetallic NPs



Schematics of MD arrangement - correspondence to experimental setup

Nucleation kinetics

MD growth of monometallic NPs



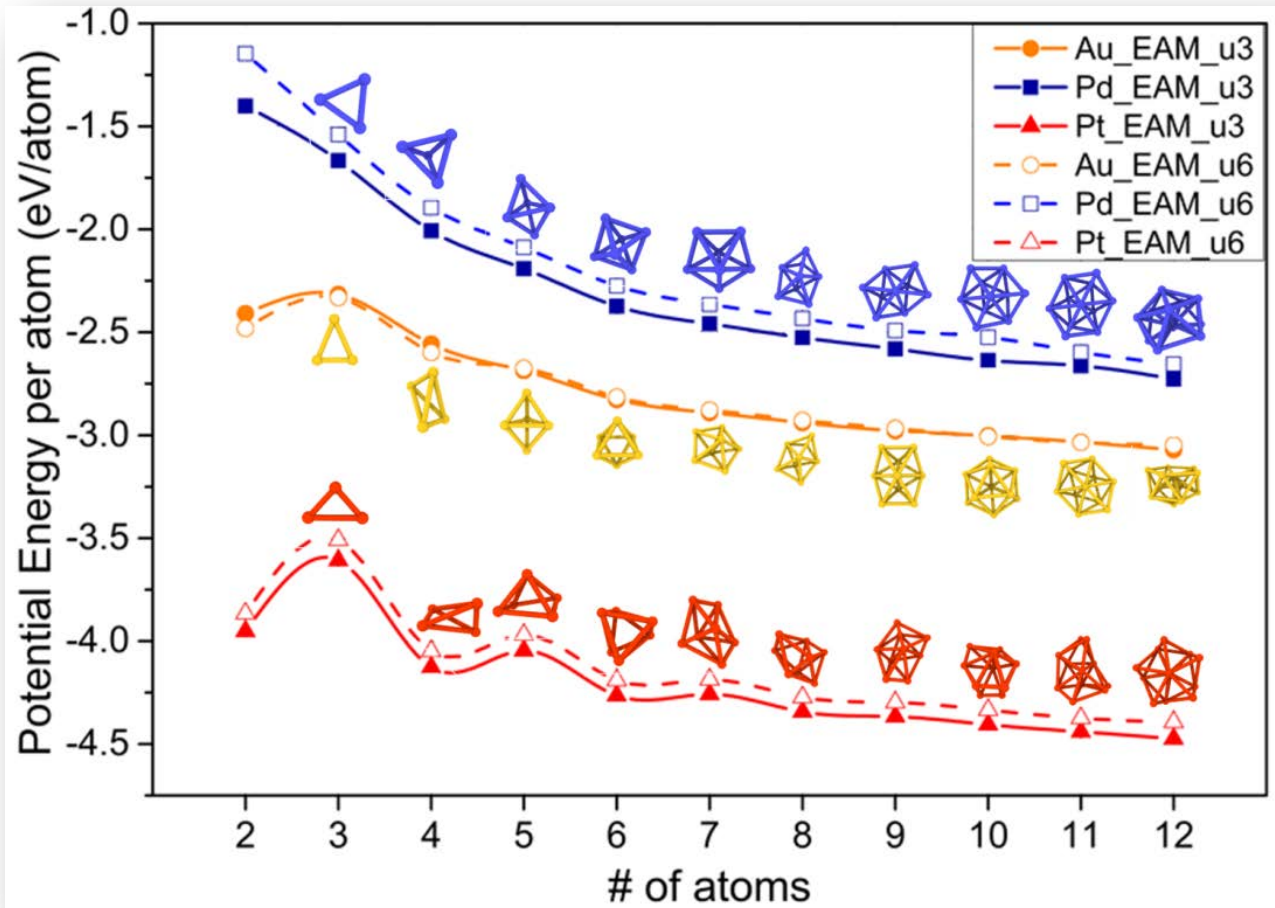
Growth enhanced by low temperature and high magnetron power

Pd NPs form faster than Au or Pt

Temperature **spikes** for Au and Pt correspond to nucleation of **dimers**, not immediately followed by further growth

Pd clusters of **larger sizes** readily form: high and **broad** temperature peak

Nucleation kinetics



Potential energy per atom as a function of number of atoms for low nuclearity monometallic clusters

Two EAM-type potentials: **Enhanced stability for Au and Pt dimers compared to trimers**

Stability of Pd clusters increases monotonically with size

NP growth

		surface view	atom type	atomic distribution	crystalline phase
(1)	Au-rich 20hedron				
(2)	Au-rich 20hedron				
(3)	small Au-rich 20hedron				
(4)	small Au-rich 20hedron				
(5)	Pd-rich 20hedron				
(6)	Pd-rich 20hedron				
(7)	Pd-rich T-8hedron				
(8)	Pd-rich T-8hedron				
(9)	Pd-rich spherical				
(10)	Au/Pt/Au T-8hedron				
(11)	large Au/Pt/Au 20hedron				

NP structures
obtained by MD growth

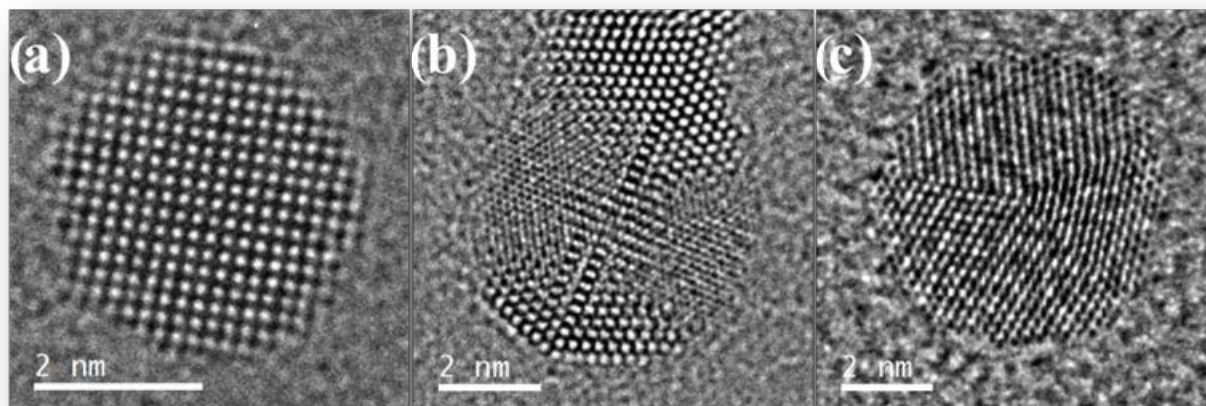
	All	Au	Pt	Pd
Ih				
Dh				
T-Oh				
sphere				

Equilibrium states of four selected geometries and
compositions by MMC

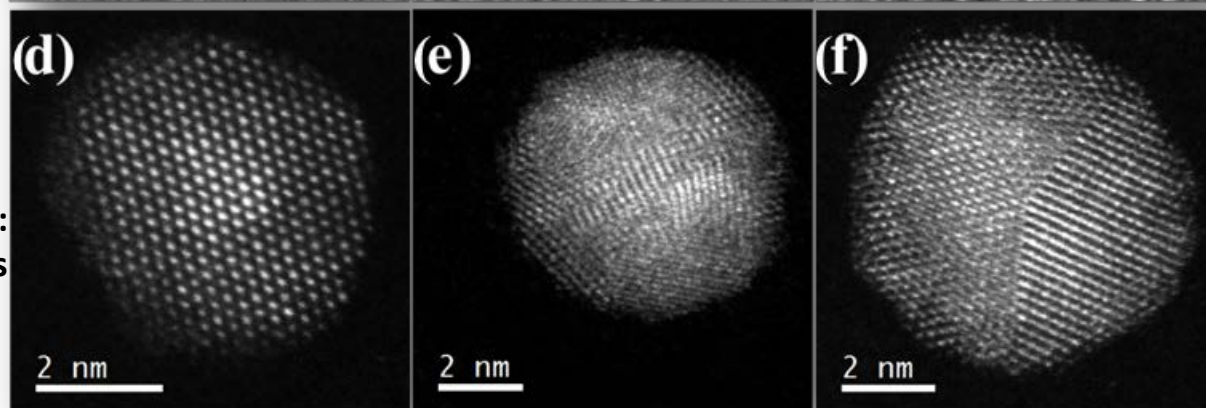
TEM characterisation

trimetallic Pd-15 W/Au-5 W/Pt-5 W sample

Physical order:
HR-TEM micrographs



Chemical order:
STEM-HAADF micrographs



(a) fcc particle viewed along its [100] zone axis

(b) icosahedral particles viewed along their twofold symmetry axes

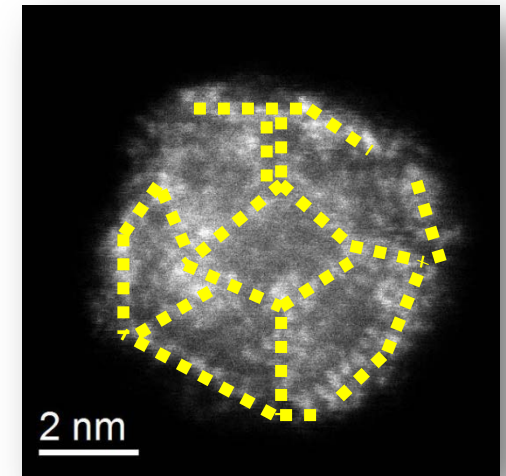
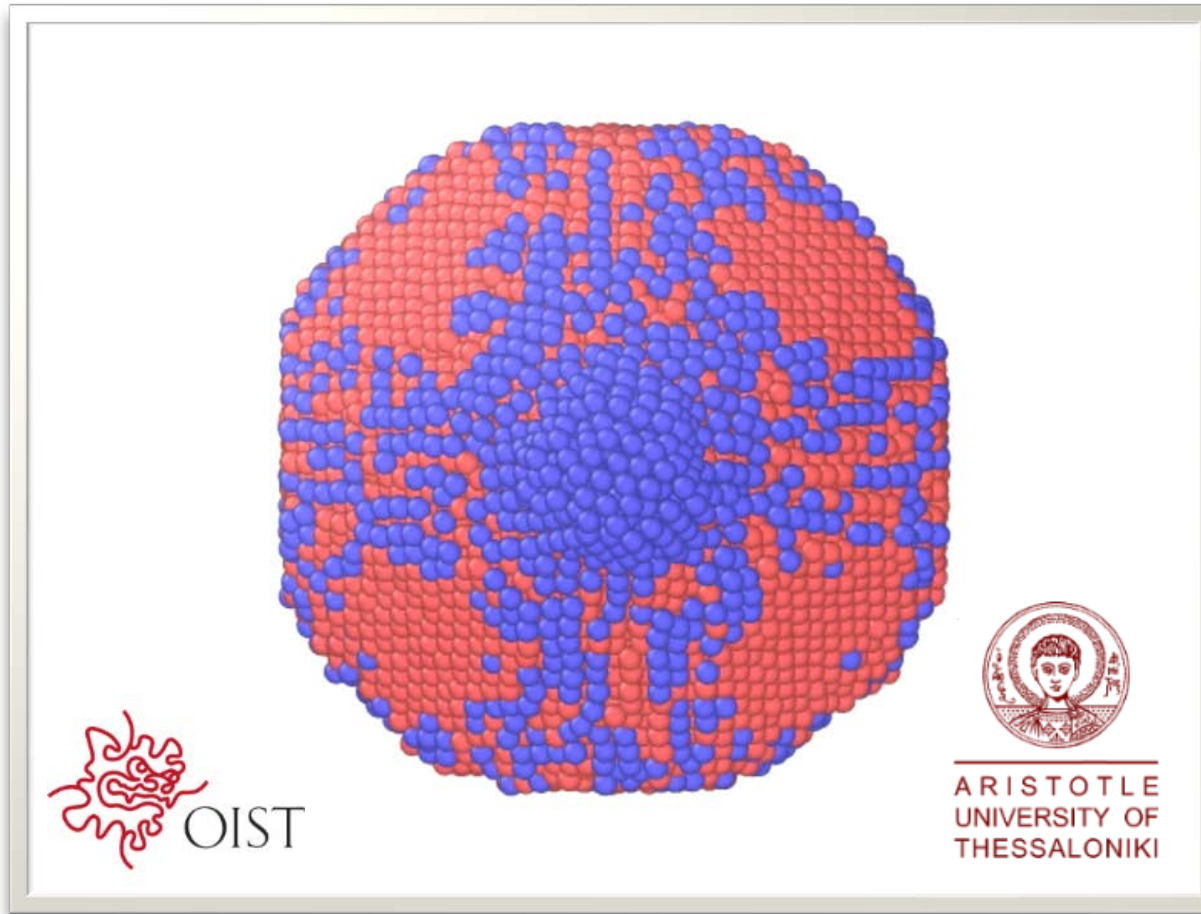
(c) decahedral particle viewed along its fivefold symmetry axis

(d) fcc particle viewed along its [101] zone axis

(e) Icosahedron viewed along its twofold symmetry axis

(f) fivefold twinned nanoparticle (decahedron)

Nanoparticle coalescence



(NiPt)

Grammatikopoulos *et al.*, Kinetic trapping through coalescence and the formation of patterned Ag-Cu nanoparticles, *Nanoscale* 8 (2016) 9780-9790

Multiple mechanisms simultaneously at play

Overview of multi-step Fe-Au nanocube growth model

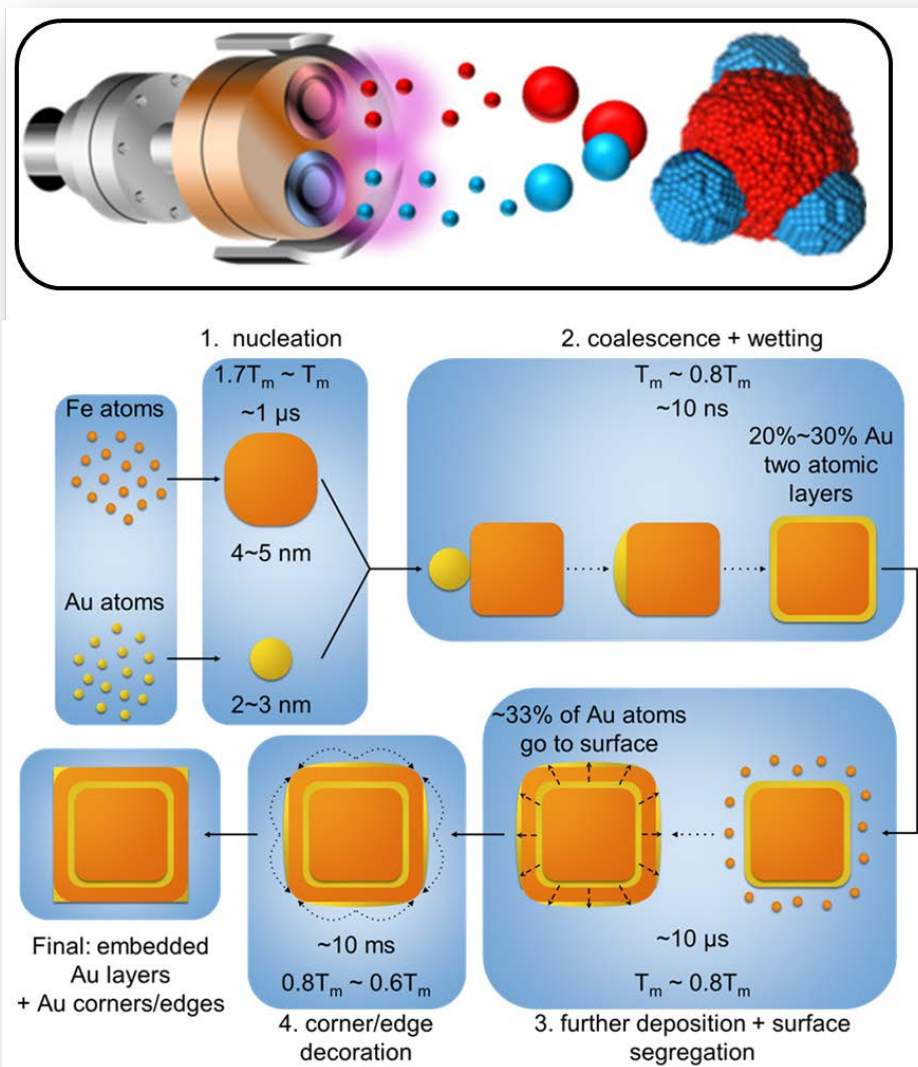
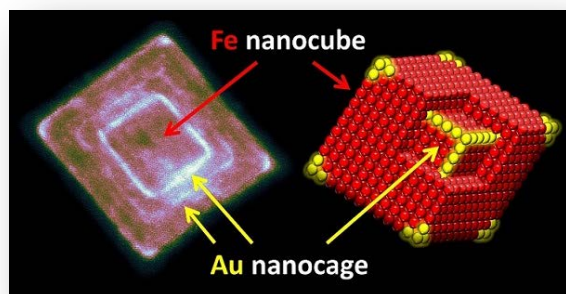
(1) Nucleation of monometallic clusters

(2) Coalescence and wetting

(3) Concurrent further deposition of residual Fe atoms in vapour phase and surface segregation of Au atoms in the nanocube create the core-frame morphology

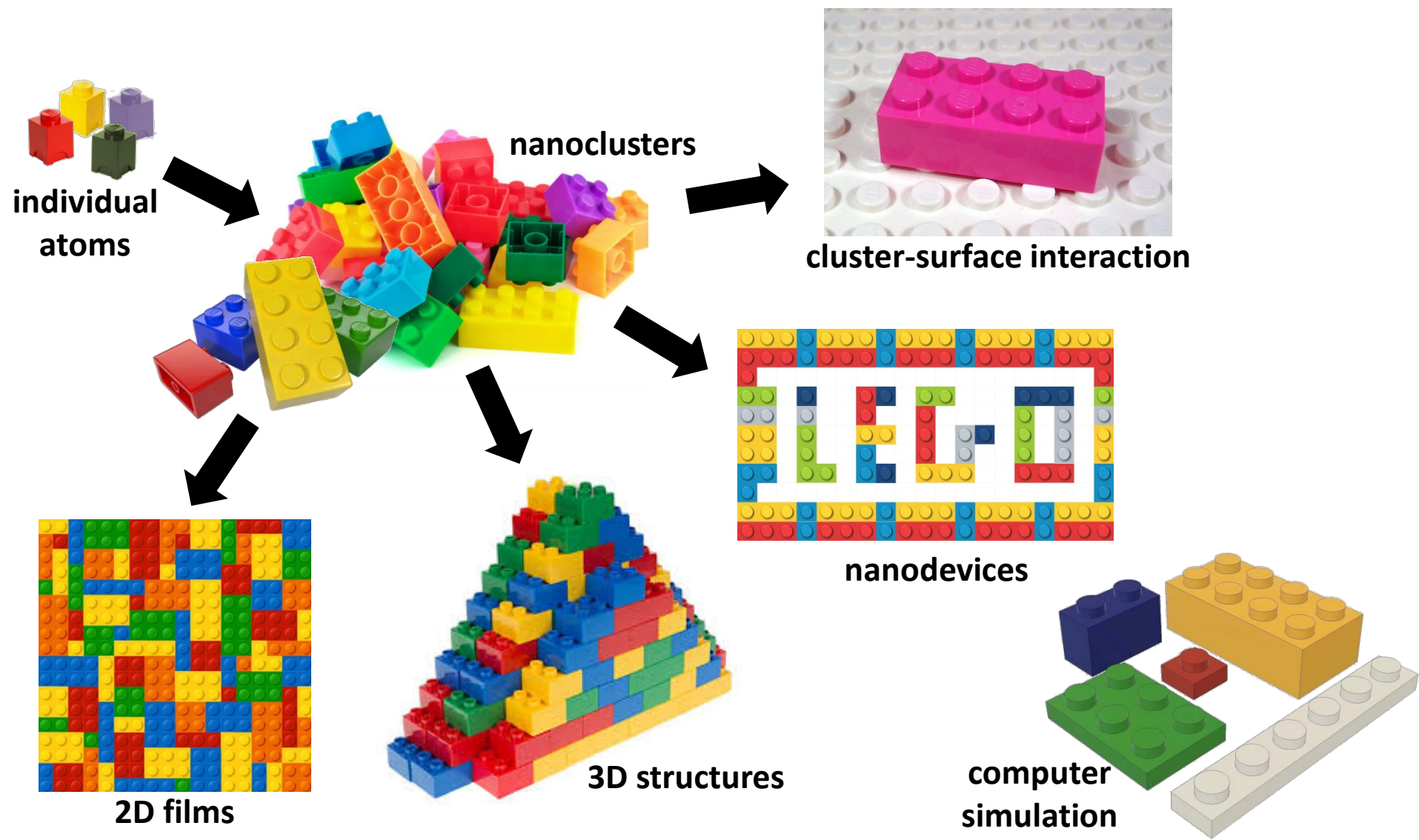
(4) Surface diffusion of Au atoms leads to vertex and edge decoration

(5) Final multiple-frame morphology.



From Nanoparticles by Design...

... to Design by Nanoparticles

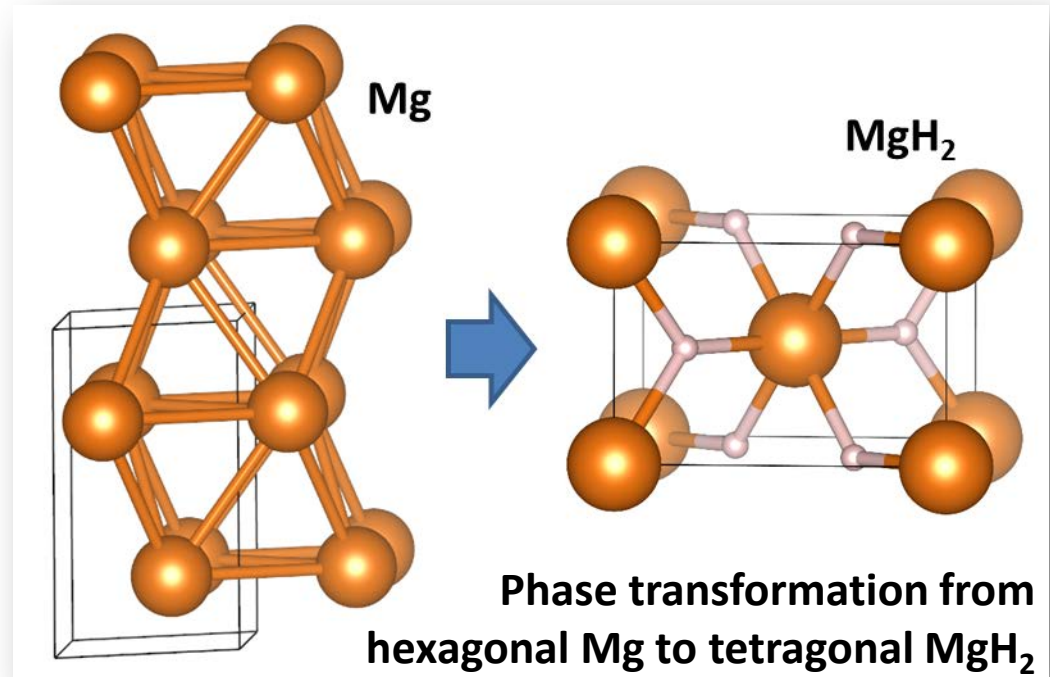


Nanoportals for H-storage and Electrocatalysis

- Mg is an attractive material for H storage due to its high reversible H mass capacity of 7.6 wt.%

Problems

- High desorption temperature
- Slow room-temperature hydrogen sorption kinetics

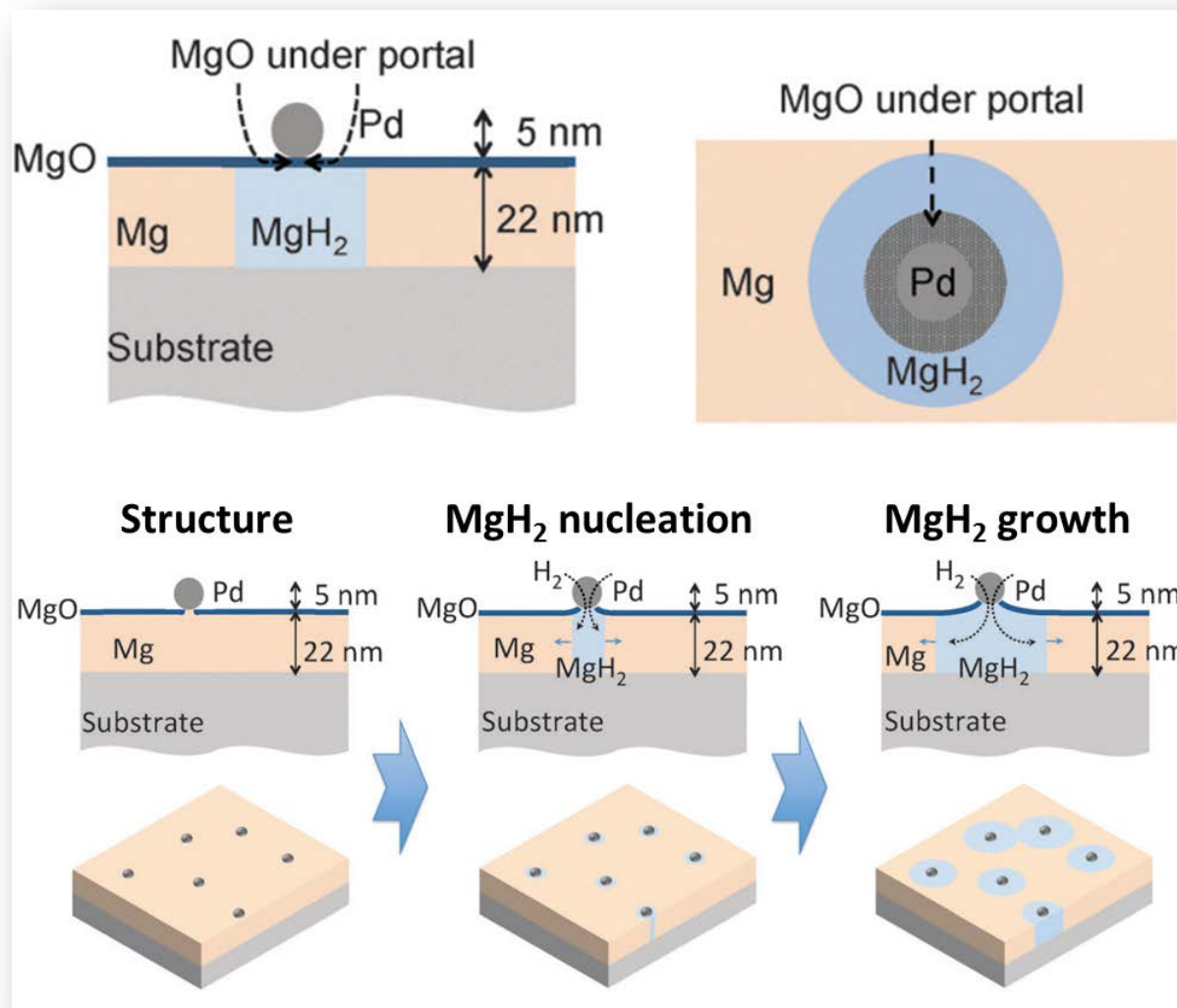


Use of Pd capping layers the answer?

Pd-capped Mg nanofilms suffer from a “**blocking-effect**” of MgH₂ at the Pd–Mg interface:

- use of high temperatures in order to accelerate the kinetics results in Pd inter-mixing with Mg, which typically **forms intermetallic compounds** (Mg₅Pd₂, Mg₃Pd, Mg₆Pd)
- the Pd capping layer **often delaminates** after a few sorption cycles, resulting in reduced MgH₂ formation

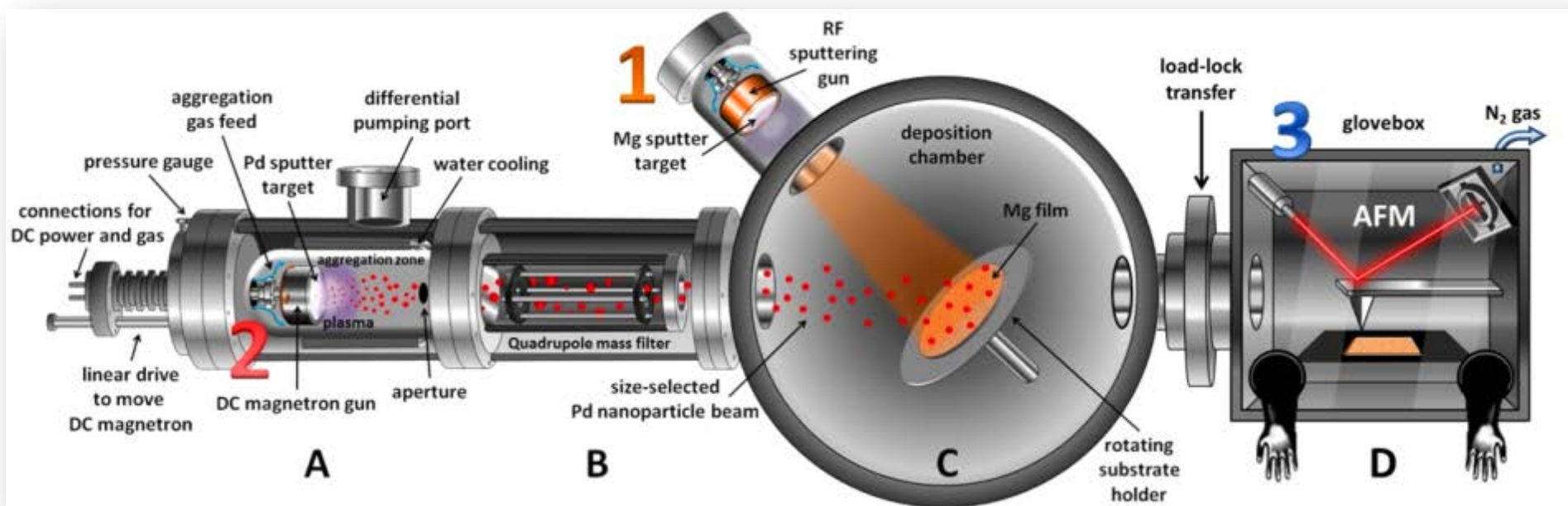
Advantages of using nanoparticles



- ✓ NPs are generally well-known to exhibit enhanced catalytic activity, compared to bulk or nanofilm systems
- ✓ compressive stress and delamination can be prevented
- ✓ monodispersed Pd NPs can potentially reduce the degree of alloy formation at high temperature (less Pd in contact with Mg)
- ✓ much less material used

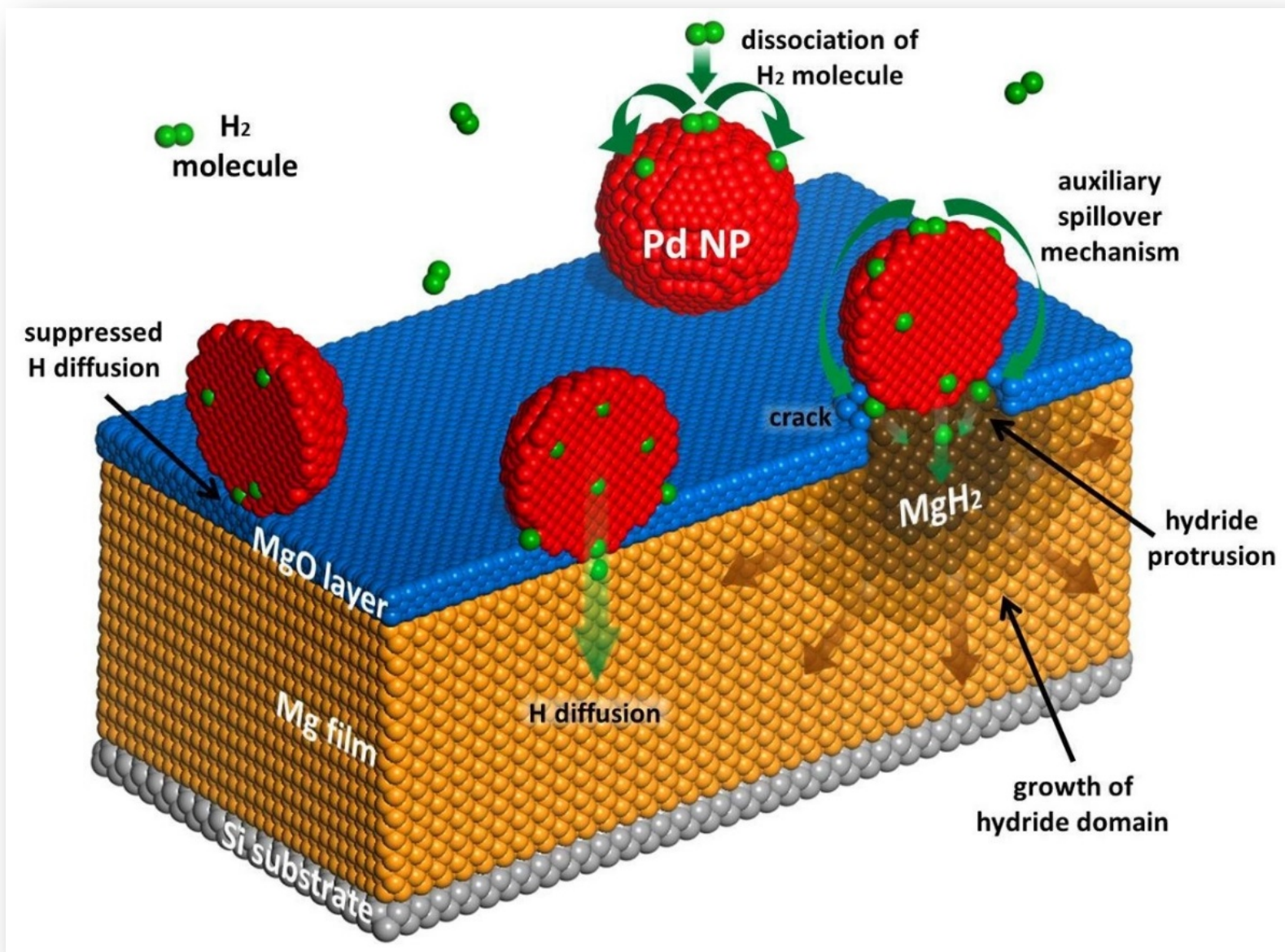
Deposition procedure

Advantage: possibility to let oxidise

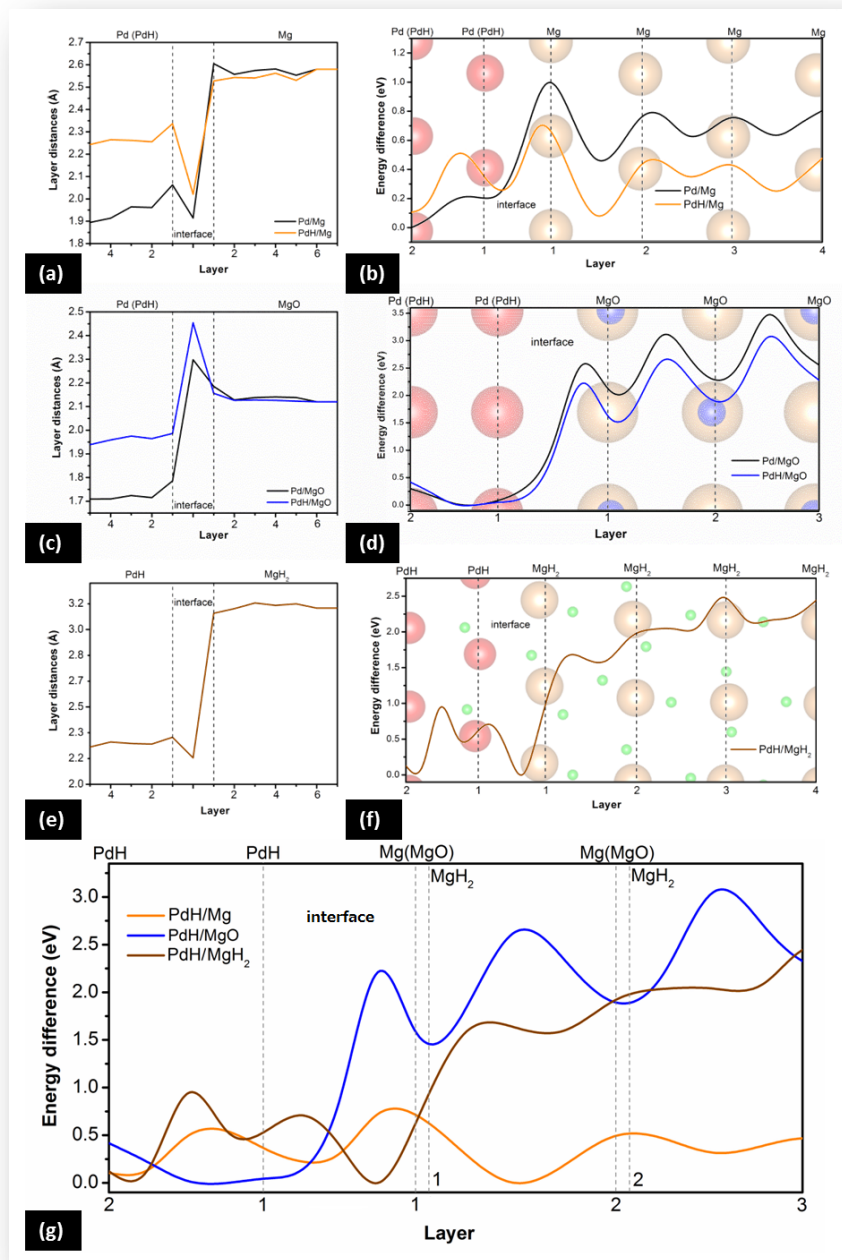
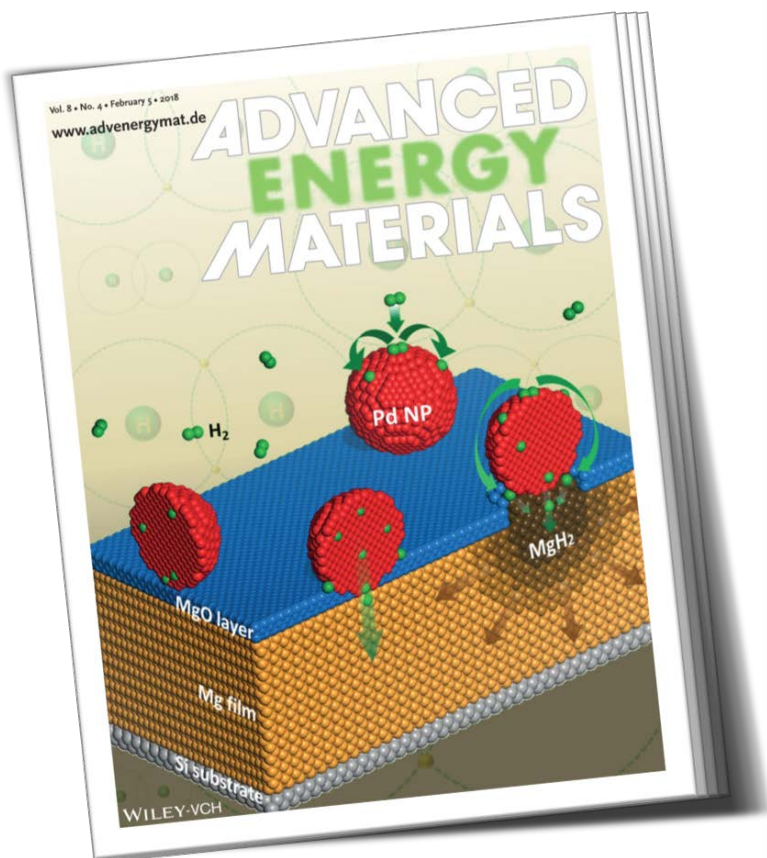


Experimental Setup

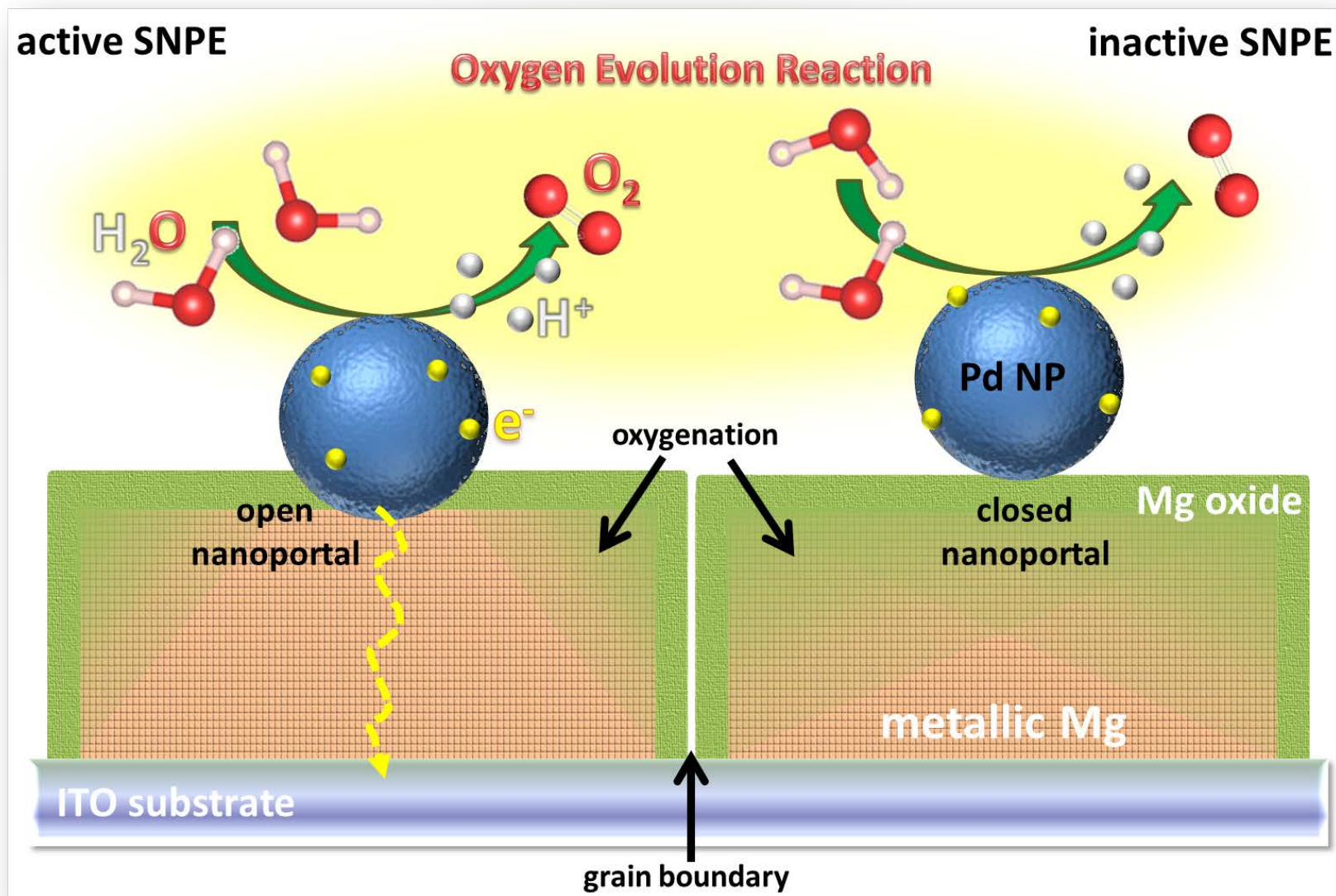
Summary of mechanism



Proof via DFT

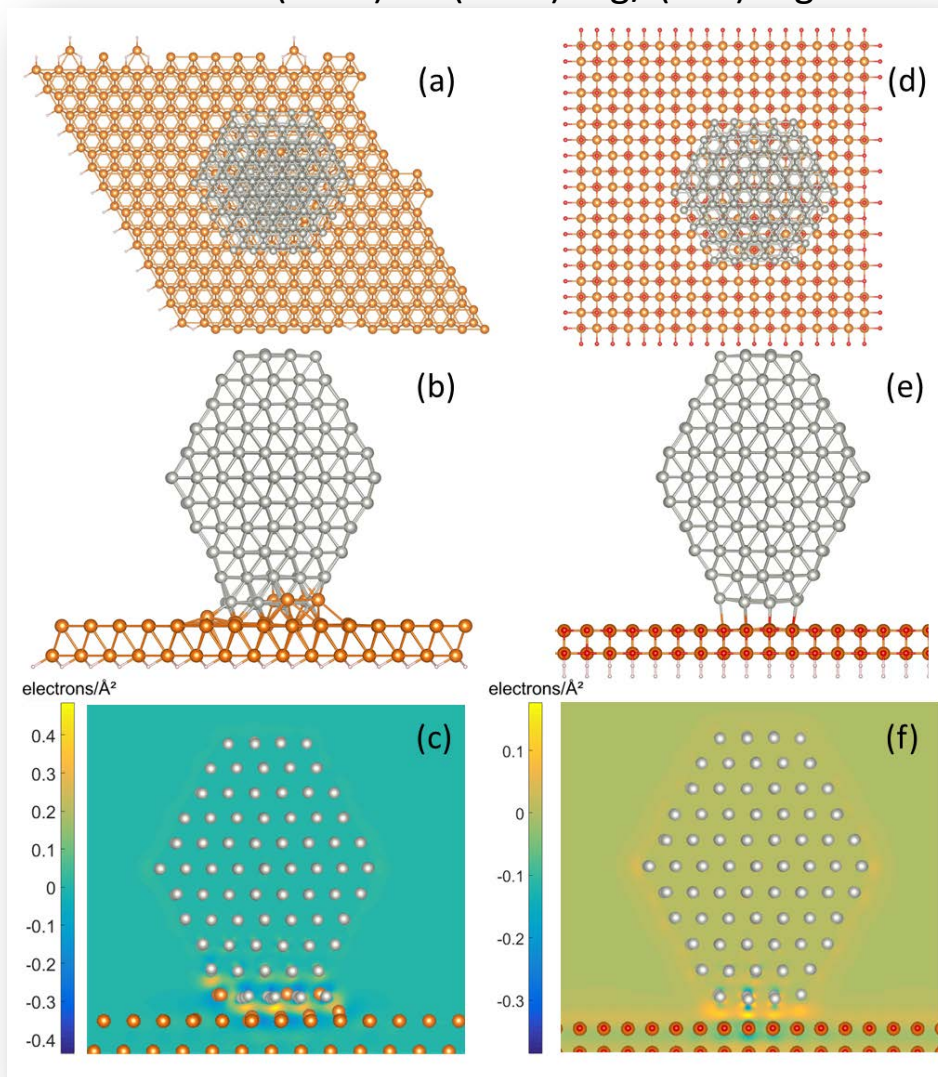


Pd nanoportals for single NP electrochemistry

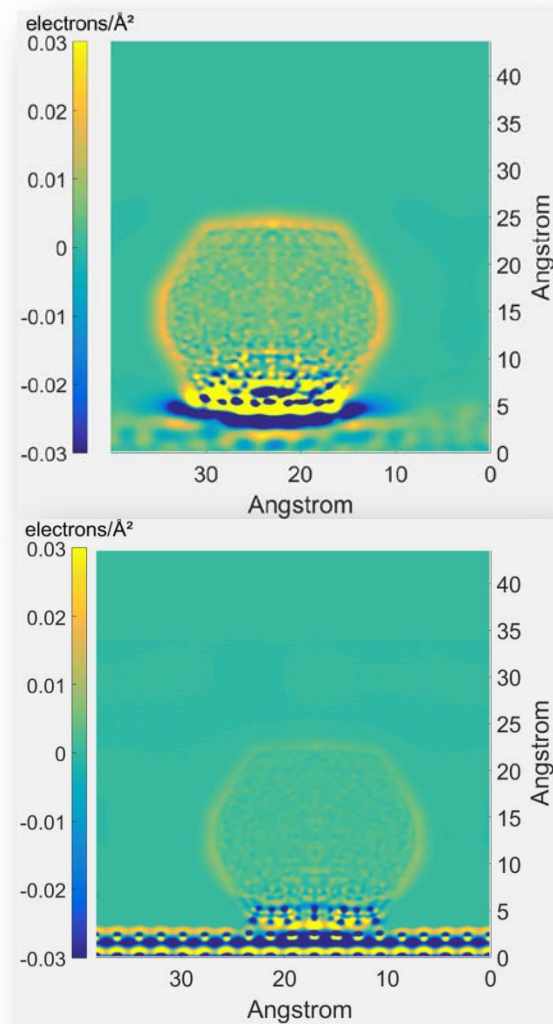


Pd nanoportals for single NP electrochemistry

HCP Pd NP (0001) on (0001) Mg/ (100) MgO



20hedral Pd NP (111) on (0001) Mg/ (100) MgO



**Thank you for your
attention!**